

**STUDY GUIDE FOR COMPREHENSIVE EXAM
MATERIALS PROCESSING
2010**

MECH 370 – Principles of Materials Processing

Text Book – Phase Transformations in Metal and Alloys, 2nd. or 3rd. Edition, D.A. Porter
and K.E. Easterling (Chapman & Hall, or Nelson Thornes editions)

Sections to be Covered:

- I. Thermodynamics (All Chapter 1, except 1.7)
 - a. Gibb's Free Energy and Chemical Potential, 1.1-1.2
 - b. Binary Solid Solutions, 1.3.1-1.3.5, 1.3.7-1.3.8
 - c. Real Solutions, 1.3.6
 - d. Equilibrium Binary Phase Diagrams, 1.5
 - e. Interfaces, 1.6, 3.1
 - i. Surface Tension, 3.1
 - ii. Gibb's Thomson Effect, 1.6
 - a. Eyring Rate Theory, 1.9
 - b. Chemical Potential as a Driving Force, 2.4
 - c. Continuity Equation, 2.4
 - d. Atomistic Models, 2.1
 - i. Self, 2.3.1
 - ii. Interstitial, 2.2
 - iii. Substitutional, 2.3
 - e. Diffusion Paths, 2.7
 - a. Nucleation & Growth 4.1-4.2
 - i. Homogeneous
 - ii. Heterogeneous
 - a. Nucleation
 - i. Homogeneous, 5.1
 - ii. Heterogeneous, 5.2
 - b. Growth
 - i. Interfaces, 3.4
 - ii. Precipitate Growth, 5.3
 - iii. Kinetics (JMA) 5.4
 - c. Recovery, Recrystallization and Grain Growth
 - i. Descriptions of Grain Boundaries, 3.3.1-3.3.2
 - ii. Driving Force for, 3.3.3-3.3.4
 - iii. Zener Pinning, 3.3.5
 - d. Age Hardening, 5.5.1-5.5.4
 - e. Spinodal Decomposition, 5.5.5
- II. Diffusion (All Chapter 2, except 2.6 and 2.8)
- III. Solidification (4.1 and 4.2)
- IV. Diffusional Transformations (Chapter 5)

- f. $\gamma \rightarrow \alpha$ and other
 - i. ferrite to austenite, 5.6
 - ii. eutectoid reaction, 5.8
 - iii. massive transformations, 5.9
 - iv. ordering transformations, 5.10
 - a. Characteristics of, 6.1
 - b. Bain model – FCC \rightarrow BCT, 6.2
 - c. Hardenability of Steel (TTT diagram), 5.8, 6.6, 6.7.1
- V. Diffusionless Transformations (parts of Chapter 6)

Sample Short Questions:

Explain the following using simple diagrams, if needed:

- a) What the driving force is for the formation of the transition phases in a Cu-1.9wt.%Be alloy undergoing the transformation: **supersaturated solid solution** \rightarrow **G.P. zones** \rightarrow γ' \rightarrow γ .
- b) Recrystallization behaviour was observed to follow the J-M-A relationship $f(t) = 1 - \exp(-kt^n)$, where $f(t)$ is the fraction recrystallized. Would you expect n , or k to be temperature dependant? Explain.
- c) Why the grain boundary diffusion rate is higher than bulk (volume) diffusion rate at lower temperatures, and the reverse at higher temperatures.
- d) Why the self-diffusivity of pure Al rapidly quenched from 600°C is higher than the steady-state value by a factor of $10^8 - 10^9$.
- e) Why dendrites form when a solid grows in a liquid (or when snowflakes form).
- f) Why ΔG_v for solidification increases with increasing undercooling (ΔT) below the equilibrium melting temperature (T_m).
- g) Why an undercooling (ΔT) is required for homogeneous nucleation of a second phase.
- h) Why the equilibrium solubility of a dispersed phase in a 2-phase mixture increases as the particle size decreases.
- i) Why the nucleation rate for phase transformations exhibits 'C-curve kinetics' (i.e., first increases with decreasing temperature, then decreases).
- j) Why heterogeneous nucleation (on g.b.'s, dislocations, pre-existing, phases...) is the dominant nucleation mechanism in solid-solid phase transformations.
- k) Why a diffusionless (martensite) transformation is called a 'shear' transformation.
- l) How the Continuous-Cooling-Transformation (CCT) diagram for steel can be used to determine its hardenability.
- m) How a dispersion of stable second-phase precipitates can prevent grain growth during high temperature annealing.